



# Effect of exfoliation and surface deposition of $\text{MnO}_x$ species in $\text{g-C}_3\text{N}_4$ : Toluene photo-degradation under UV and visible light



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## ABSTRACT

A relatively low surface area graphitic carbon nitride produced by a standard melamine calcination procedure was subjected to an exfoliation process and/or surface deposition of Mn-containing species. The resulting catalysts were characterized using X-ray diffraction and photoelectron spectroscopies, transmission electron microscopy, UV–vis and infrared spectroscopies, and porosimetry. The photocatalytic properties of the materials were tested in the gas-phase degradation of toluene under UV, visible and Sunlight-type illumination and their performance assessed through the measurement of the materials optical properties and subsequent calculation of the true quantum efficiency. The numerical analysis of the efficiency parameter evolution throughout the above mentioned carbon nitride samples provides quantitative estimation of the catalytic effect resulting from the exfoliation process and/or the surface deposition of Mn-containing species. Such analysis allows thus to compare the activity significance of each one of the carbon nitride modifications explored as well as the synergistic effect originated by the combined action of such modifications.

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## 1. Introduction

The high levels of pollution originated by industrial and mobile source emissions, together with the ever-higher standards for emission control of many industrialized countries, have driven the current interest in environmental protection processes, leading to a high scientific and technological activity in this area. In this context, much attention has been focused on advanced oxidation processes and in particular heterogeneous photocatalysis. Heterogeneous photocatalysis is an advanced oxidation process that uses semiconductors, broadly applied in the degradation/transformation of organic pollutants as well as biological microorganisms [1–4]. Its most significant advantage with respect to other remediation technologies could be the use of the solar energy, a free and renewable energy source. Obviously, the efficiency of this process depends directly of the photocatalyst performance upon solar irradiation conditions.

Undoubtedly,  $\text{TiO}_2$  has been the most studied photocatalyst. For such material, many studies reported relatively high activity. Addi-

tionally, its low cost and stability makes this material particularly interesting for a wide variety of applications [1,5–8]. However, the limited utilization of the visible and infrared regions of the solar spectrum precludes its wide potential application as a solar photocatalyst. Such limitation comes from the high band gap inherent by all titania polymorphs, around 3.0–3.5 eV [1,7,9–13]. Thus, and despite the significant use of  $\text{TiO}_2$ -based heterogeneous photocatalysts, it is especially important the development of new, alternative semiconductors focusing in the effective utilization of the sunlight. This involves unavoidably the improvement of the visible light use as this region accounts for ca. 43% of the intensity of the solar spectrum.

One of the materials potentially interesting for sunlight-driven photocatalytic applications is the graphite like carbon nitride ( $\text{g-C}_3\text{N}_4$ ). Its good electronic and chemical properties and thermal stability have allowed the use of the  $\text{g-C}_3\text{N}_4$  material as a free metal catalysis for many photochemical reactions such as organic photodegradation, water splitting or alcohol reforming [14–19]. In order to optimize the carbon nitride catalytic properties for target-specific applications, several modifications of bare  $\text{g-C}_3\text{N}_4$  have been conducted, including the surface modification using metals such as Ag, Au, Pt [20–30], and different oxides or hydroxides, among which we can highlight  $\text{ZnO}_2$ ,  $\text{TiO}_2$ ,  $\text{WO}_3$ ,  $\text{Cu}_2\text{O}$ ,  $\text{Co}(\text{OH})_2$  and  $\text{MnO}_x\text{OH}_y$  [31–41]. Another possible modification consists

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in obtaining exfoliated systems. This greatly increases the surface area of the g-C<sub>3</sub>N<sub>4</sub> bulk. The limited surface area presented by the bulk material, typically 10–40 m<sup>2</sup>g<sup>−1</sup>, is a consequence of the stacking of the g-C<sub>3</sub>N<sub>4</sub> layers and can be significantly increased by the mentioned exfoliation process [15,20,30,42,43]. Moreover, many authors report that exfoliated g-C<sub>3</sub>N<sub>4</sub> nanomaterials can modify positively the photo-charge handling capability of the material with respect to the bulk counterpart [16,44–46]. There are several methods of exfoliating g-C<sub>3</sub>N<sub>4</sub>. One of the simplest is the ultra-sonication assisted by water. In this case, the exfoliated g-C<sub>3</sub>N<sub>4</sub> material presents high stability and lacks toxic impurities (common by other preparation methods), which make them a promising material for photocatalytic processes [44].

In this contribution we combine two strategies related to the exfoliation and surface modification by Mn-containing species to obtain a new material which enhanced catalytic properties with respect to the bare, bulk-type g-C<sub>3</sub>N<sub>4</sub>. Mn-containing species have attracted particular attention due to their significant influence in photoactivity through the efficient handling of hole-related species while exfoliation appears, as mentioned, as a general way to improve photo-performance [39]. We tested such materials in the photodegradation of gas phase toluene. In order to fully analyze the potential of the materials in utilizing efficiently the sun as the energy source of the photodegradation process, the study was carried out using three light sources, this is; UV, sunlight-type and visible type. The characterization of the samples was performed using X-ray diffraction, X-ray photoelectron, infrared, and ultraviolet spectroscopy, transmission electron microscopy and N<sub>2</sub> physisorption. Additionally, the study includes the modelling of the light-matter interaction, necessary for determining the true quantum efficiency values and compare samples in quantitative basis. Using the true quantum efficiency parameter and a simple mathematical procedure we carried out a quantitative analysis of the beneficial effects induced by the exfoliation process of the carbon nitride and/or the presence of Mn-containing species in the photoactivity of the materials.

## 2. Materials and methods

### 2.1. Samples preparation

The graphitic carbon nitride was obtained by calcination of melamine (Aldrich), in a semi-closed system to prevent sublimation, at 580 °C for 4 h using a heating ramp of 5 °C min<sup>−1</sup> [20,39,47]. The exfoliated g-C<sub>3</sub>N<sub>4</sub> material was prepared by ultrasonication for 4 h using as-prepared bulk g-C<sub>3</sub>N<sub>4</sub> in deionized water (1 mg cm<sup>−3</sup>). The suspended part after 18 h of sedimentation was selected as the exfoliated part of the material and obtained after evaporation of the water at 80 °C. The carbon nitride component modified with manganese oxide was prepared using an impregnation method. For this, the g-C<sub>3</sub>N<sub>4</sub> or the exfoliated one and the manganese precursor (Manganese (II) nitrate tetrahydrate, Aldrich) were added to a deionized water solution. After 2 h under magnetic stirring the liquid was evaporated at 110 °C for 15 h. The final MnO<sub>x</sub>-carbon nitride composites were obtained by calcination at 350 °C, with a heating ramp of 5 °C min<sup>−1</sup> maintaining this temperature for 2 h. The final MnO<sub>x</sub> loading was 2 wt.%, confirmed with an error below 5% using ICP-AAS (induced couple plasma and atomic absorption spectroscopy) spectrometry.

Samples names are g-C<sub>3</sub>N<sub>4</sub> or g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub> when prepared from the initial powder extracted from calcination of melamine and E-g-C<sub>3</sub>N<sub>4</sub> or E-g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub> when prepared from the powders subjected to the exfoliation procedure.

### 2.2. Characterization and analytical methods

The BET surface areas and average pore volumes and sizes were measured by nitrogen physisorption (Micromeritics ASAP 2010). XRD profiles were obtained using a Seifert D-500 diffractometer using Ni-filtered Cu Kα radiation with a 0.02° step. UV-vis diffuse-reflectance spectroscopy experiments were performed on a Shimadzu UV2100 apparatus using nylon as a reference. The Fourier transform infrared spectra were taken in a Bruker Vertex 80 FTIR spectrometer using a MCT detector. XPS data were recorded on 4 × 4 mm<sup>2</sup> pellets, 0.5 mm thick, prepared by slightly pressing the powdered materials which were outgassed in the prechamber of the instrument at room temperature up to a pressure <2 × 10<sup>−8</sup> Torr remove chemisorbed water from their surfaces. The SPECS spectrometer main chamber, working at a pressure <10<sup>−9</sup> Torr, was equipped with a PHOIBOS 150 multichannel hemispherical electron analyser with a dual X-ray source working with Ag Kα (hν = 1486.2 eV) at 120 W, 20 mA using C 1s as energy reference (284.6 eV). Transmission electron microscopy images were recorded on a JEOL 2100F TEM/STEM microscope.

### 2.3. Photo-catalytic experimental details

Gas-phase photo-oxidation of toluene (≥99% Aldrich) was carried in a continuous flow annular photoreactor containing ca. 0.4 mg cm<sup>−2</sup> of photocatalyst as a thin layer coating on a pyrex tube. The scheme of the reaction system is presented in Fig. 1 and has been presented and detailed in previous works [48,49]. The reacting mixture (100 ml min<sup>−1</sup>) was prepared by injecting toluene into a wet (ca. 75% relative humidity) 20 vol.% O<sub>2</sub>/N<sub>2</sub> flow before entering to the photoreactor, yielding an organic inlet concentration of ca. 700 ppmv. Fluorescent UV (Sylvania F6WBLT-65; 6 W) and sunlight-type lamps (Sylvania F6W/D; 6 W) were used for the photoreaction experiment. For the visible light experiments, we used a flexible polyester filter made from a deeply dyed PET material to absorb ultraviolet rays. The material allows less than 10% transmission below 390 nm. Reaction rates and selectivity were evaluated under steady-state conditions, typically achieved after ca. 6–10 h from the irradiation starting. The concentration of toluene and the reaction products were analyzed using an on-line gas chromatograph (Agilent GC 6890) equipped with a TCD (for CO<sub>2</sub> measurement) and FID (organics measurement) detectors.

### 2.4. Light-matter interaction modelling and quantum efficiency calculation

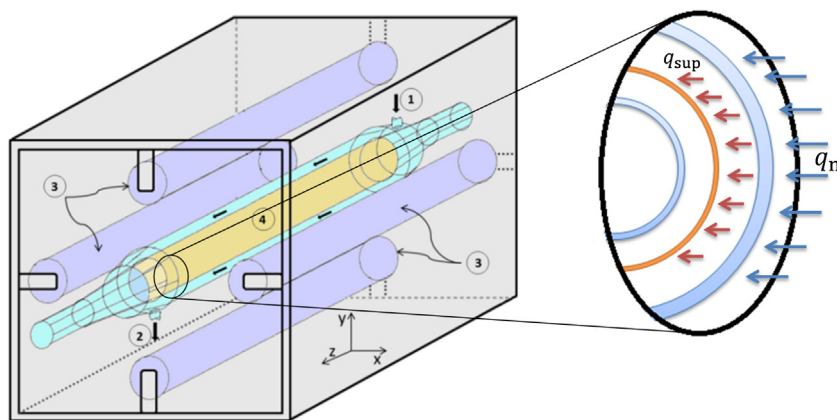
#### 2.4.1. Local superficial rate of photon absorption

The local superficial rate of photon absorption ( $e^{a,s}$ ) is defined by Eq. (1) [31,50]. In this equation  $F_{As}$  is the fraction of light absorbed by the sample and  $q_{sup}$  the radiation flux at each position ( $X_s, Y_s, Z_s$ ) of the catalytic film (See Fig. 1).

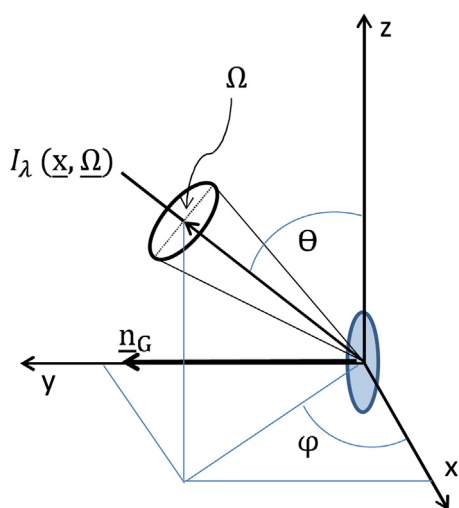
$$e^{a,s}(\underline{x}) = q_{sup}(\underline{x}) F_{As} \quad (1)$$

To obtain the radiation flux on the surface of the samples, we calculate first the impinging radiation flux from the lamps ( $q_n$  in Fig. 1). Considering the coordinated system presented in Fig. 2 and the geometry of the reactor (annular multilamp), the  $q_n$  can be determined by Eq. (2) [31].

$$q_n(X_s, Y_s, Z_s) = \sum_{L=1}^{L=4} \sum_{\lambda} \int_{\varphi_{min,L}(x,y)}^{\varphi_{max,L}(x,y)} \int_{\theta_{min}(x,y,\varphi)}^{\theta_{max}(x,y,\varphi)} \frac{P_{\lambda,L}}{2\pi R_L Z_L} \sin^2 \theta \left( \left( \frac{X_s - X_L}{R} \right) \cos \varphi + \left( \frac{Y_s}{R} \right) \sin \varphi \right) d\varphi d\theta \quad (2)$$



**Fig. 1.** Photocatalytic annular reactor scheme. (1) gas inlet, (2) gas outlet, (3) lamps, (4) catalyst sample.  $q_{sup}$  radiation flux on the surface of the sample (red),  $q_n$  radiation flux from the lamps (blue). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)



**Fig. 2.** Coordinated system. Center of coordinates located at the sample (defined by coordinates  $x_s, y_s, z_s$ ).

Where the integration limits can be evaluated using the ray tracing method and are summarized in Eqs. (3)–(10) [31,51].

$$\varphi_1 = \tan^{-1} \left( \frac{X_L - X_s}{Y_L - Y_s} \right) \quad (3)$$

$$\varphi_2 = \sin^{-1} \left( \frac{R_L}{(X_L - X_s)^2 + (Y_L - Y_s)^2} \right) \quad (4)$$

$$\varphi_{min} = \varphi_1 - \varphi_2 \quad (5)$$

$$\varphi_{max} = \varphi_1 + \varphi_2 \quad (6)$$

$$\theta_{min}(\varphi) = \cos^{-1} \frac{-Z_s}{(X_{Lm}(\varphi) - X_s)^2 + (Y_{Lm}(\varphi) - Y_s)^2 + Z_s^2} \quad (7)$$

$$\theta_{max}(\varphi) = \cos^{-1} \frac{Z_L - Z_s}{(X_{Lm}(\varphi) - X_s)^2 + (Y_{Lm}(\varphi) - Y_s)^2 + Z_s^2} \quad (8)$$

Where:

$$X_{Lm}(\varphi) = X_L + (X_s - Y_L) \cos \varphi^2 + (Y_L - Y_s)(\cos \varphi \sin \varphi) - \sin \varphi \sqrt{(R_L^2 - (X_s - X_L) \cos \varphi + (Y_L - Y_s)(\sin \varphi)^2)} \quad (9)$$

$$Y_{Lm}(\varphi) = Y_s + (Y_L - Y_s) \cos \varphi^2 + (X_s - X_L)(\cos \varphi \sin \varphi) - \cos \varphi \sqrt{(R_L^2 - (X_s - X_L) \cos \varphi + (Y_L - Y_s)(\sin \varphi)^2)} \quad (10)$$

$X_L, Y_L, Z_L$  are the coordinates of the points located on the surface of the lamp.  $X_s, Y_s$  and  $Z_s$  are the coordinates of the points located on the surface of the films.

Finally, the  $q_{sup}$  x/y components (see Figs. 1 and 2; Eq. (11)) can be determined using  $q_n$  and a radiation balance, which considers the main optical (Transmittance/Absorbance and Reflectance) events occurring in all components of the reactor placed between the emission source and catalyst, i.e. glass and reaction media, as well on the catalytic film.

$$q_{sup} = \sqrt{xq^2 + yq^2} \quad (11)$$

A detailed description of the mathematical formulation to provide  $q_{sup}$  as a function of  $q_n$  (Eq. (2)) and the transmittance/reflectance optical measurements for each component of our reactor system can be found elsewhere [31] and is summarized at the Supporting information section.

#### 2.4.2. Quantum efficiency

Quantum efficiency values ( $\eta_q$ ) were determined using Eq. (12) [7,52–57]. This work presents the determination of the true quantum efficiency using a stringent approach for calculating the light-matter interaction in the reactor (described previously in subsection 2.4.1) and including explicitly the selectivity towards the reaction products as a way to measure the number of charge species involved in the generation of each reaction product (see Eq. (12)).

$$\eta_q = \frac{r}{e^{a,s} \times S} \times 100 \quad (12)$$

Where  $r$  is the reaction rate ( $\text{mol m}^{-2} \text{s}^{-1}$ ),  $e^{a,s}$  the superficial rate of photon absorption ( $\text{Einstein m}^{-2} \text{s}^{-1}$ ; Eq. (1)) and  $S$  is the aforementioned Selectivity factor (dimensionless). This last factor can be determined using Eq. (13) [31].

$$S = \sum_i n_i S_i \quad (13)$$

Where  $i$  runs over all products of the reaction,  $S_i$  and  $n_i$  are the fractional selectivity to product  $i$ , and the inverse of the number of charge carrier species required to obtain the specific  $i$  product.

The Supporting information section summarizes the procedure to calculate the  $\eta_i$  factors.

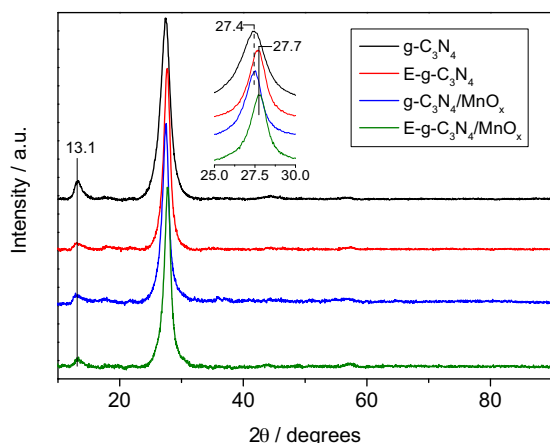


Fig. 3. XRD spectra for the carbon nitride based samples.

### 3. Results and discussion

#### 3.1. Characterization results

Morphological properties determined by  $N_2$  physisorption are summarized for all materials studied in Table 1. First of all, the specific superficial area values show differences between the bulk and exfoliated  $g\text{-C}_3\text{N}_4$  species. An increase in the BET surface area for the delaminated  $g\text{-C}_3\text{N}_4$  is detected, being larger than 290% with respect to the bulk system in all cases (i.e. in presence or absence of manganese surface species). This enhancement factor of the surface area is in line with those reported in the literature [14–19]. In addition, a modest decrease in the surface area parameter is detected when the Mn component is included in both carbon nitride samples. Table 1 also shows pore volume and pore size values of the photocatalysts. The values of these observables present a similar behavior to those of the BET surface area, this is, in all cases we observe a somewhat modified, lower porosity for samples containing  $\text{MnO}_x$  species. These results suggest that  $\text{MnO}_x$  particles tend to be located in the porous structure of the  $g\text{-C}_3\text{N}_4$ , occluding it partially.

The X-ray diffraction patterns for the bulk and exfoliated carbon nitride samples are shown in Fig. 3. All samples display the typical spectra of the graphitic  $g\text{-C}_3\text{N}_4$  layered structure, dominated by the interlayer-stacking (002) reflection. The (100) reflection, related to the interlayer structural packing motif of tri-s-triazine units, is also detected clearly in all samples at around  $13.1^\circ$  [20]. It is important to note that using this technique; it has not been possible to obtain information about the  $\text{MnO}_x$  minority phase. Finally, a  $0.3^\circ$  shift is detected in the (002) reflection for the exfoliated vs. the bulk samples, as a token related to the interlayer distance decrease taking place during the exfoliation process. It has been hypothesized that the limitation of the number of sheets stacked produces the weakening of interlayer forces with effect in the corrugation of the layer(s) and the subsequent decrease of the corresponding distance between layers [45].

Optical properties are, obviously, especially important for photocatalytic applications. Here we examined such properties using UV–visible spectroscopy. In Fig. 4, we present the spectra of the samples; they display a profile dominated by the  $g\text{-C}_3\text{N}_4$  component, leading to band gap values reported in the last column of Table 1. These values are calculated considering the materials indirect band gap semiconductors [15]. The Fig. 4 zoom view highlights the typical blue shift observed for exfoliated  $g\text{-C}_3\text{N}_4$  samples [45], which entails a band gap increment of ca. 0.1 eV (Table 1). This variation can be ascribed to the quantum confinement effect derived

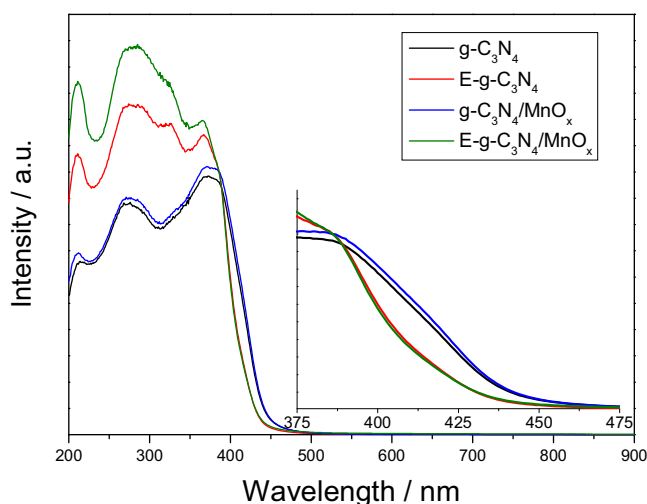


Fig. 4. UV–vis spectra for the carbon nitride based samples.

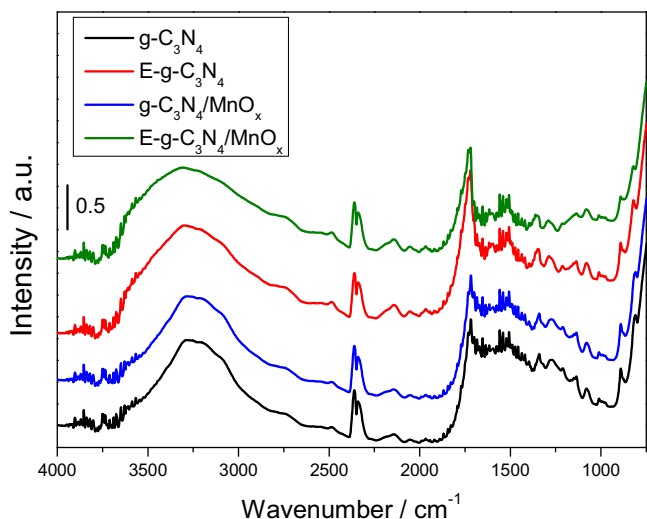


Fig. 5. FTIR spectra for the carbon nitride based samples.

from the limitation on the stacking of CN-containing layers in the  $g\text{-C}_3\text{N}_4$  nanosheet-like structure [58].

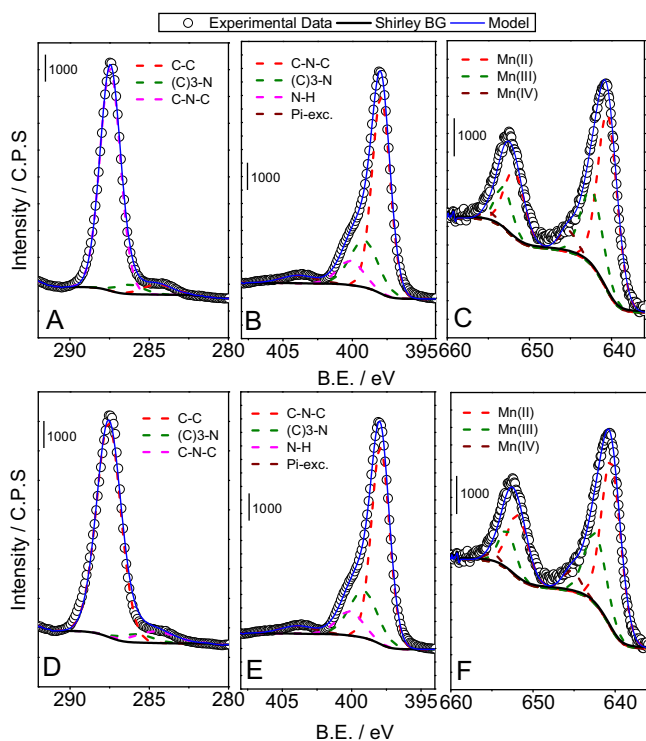
Infrared spectra of the samples obtained at RT condition are shown in Fig. 5. The N–H stretching vibration contributions at the  $3500\text{--}2500\text{ cm}^{-1}$  region are present in all samples [43,58]. In this region we have an additional contribution coming from adsorbed water molecules. The shape of this broadband shows small differences between the bulk and exfoliated samples. In the case of the exfoliated one, the band seems to be wider which could be related with an increase of the water adsorption on the surface. In the  $1600\text{--}1200\text{ cm}^{-1}$  regions we can observe several contributions mostly associated to stretching modes of CN heterocyclic moieties [59–62]. Specific differences between bulk and exfoliated samples are mostly detected around  $1700\text{ cm}^{-1}$  and may reinforce the idea that exfoliated materials have a more water-prone, hydrophilic surface. Structural differences in the C–N network can be additionally invoked to justify the differences observed between the exfoliated and bulk materials. Finally, around  $850\text{--}800\text{ cm}^{-1}$  we note additional contributions coming from the breathing modes of tri-s-triazine, which confirms that all samples maintain the  $g\text{-C}_3\text{N}_4$  block structure within the CN-containing layers, in spite of the differences in the dimension of their layer stacking [43,45,58,63].



**Table 1**  
Morphological and Optical properties for the carbon nitride based samples.<sup>a</sup>

Sample	BET Surface area (m <sup>2</sup> g <sup>-1</sup> )	Pore volume (cm <sup>3</sup> g <sup>-1</sup> )	Pore size (nm)	Band Gap (eV)
g-C <sub>3</sub> N <sub>4</sub>	21.8	0.103	15.1	2.7
E-g-C <sub>3</sub> N <sub>4</sub>	63.4	0.210	15.7	2.8
g-C <sub>3</sub> N <sub>4</sub> /MnO <sub>x</sub>	17.9	0.074	15.7	2.7
E-g-C <sub>3</sub> N <sub>4</sub> /MnO <sub>x</sub>	52.3	0.170	16.2	2.8

<sup>a</sup> Standard error: BET, 1.5 m<sup>2</sup> g<sup>-1</sup>; porosity, 8%.



**Fig. 6.** Cs (A), N1s (B) and Mn 2p (C) XPS spectra for the g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub> sample. C1s (D), N1s (E) and Mn 2p (F) XPS spectra for the E-g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub> sample. Fitting results display the different contributions to each XPS peak.

The structural analysis of the materials is completed with the help of the X-ray photoelectron spectroscopy. Fig. 6 displays XPS data for the two samples containing manganese and concerning the carbon 1s (C1s) nitrogen 1s (N1s), and manganese 2p (Mn2p) experimental regions and fitting results. No significant differences were observed between the corresponding bulk or exfoliated samples in presence or absence of MnO<sub>x</sub> species. A full summary of the fitting outcome is presented at the Supporting information section Tables S2 (for C1s) and S3 (for N1s). In Table S3 we also included information concerning the C/N atomic ratio. For C1s we detected the C–C (which has a contribution for spurious entities besides that of the main component of the sample) signal as well as those contributions related bridging carbons between aromatic moieties (C<sub>3</sub>–N) or at the aromatic rings (N–C–N). For N1s the fitting takes into account the last two species mentioned as well as the contribution from N–H moieties [15,64–66]. The fitting results summarized in Tables S2 and S3 provide evidence of the strong similitude detected between all samples. A mild difference is measured in the C/N ratio, with an increase for exfoliated vs. bulk samples, as previously detected by other authors as the stacking of carbon nitride layers diminished [64,65].

The XPS study also renders information about the MnO<sub>x</sub> component. The fitting indicate the presence of a mixture of Mn(II), Mn(III) and Mn(IV) states in both bulk and exfoliated carbon nitride samples where the Mn component is present (see results at Table

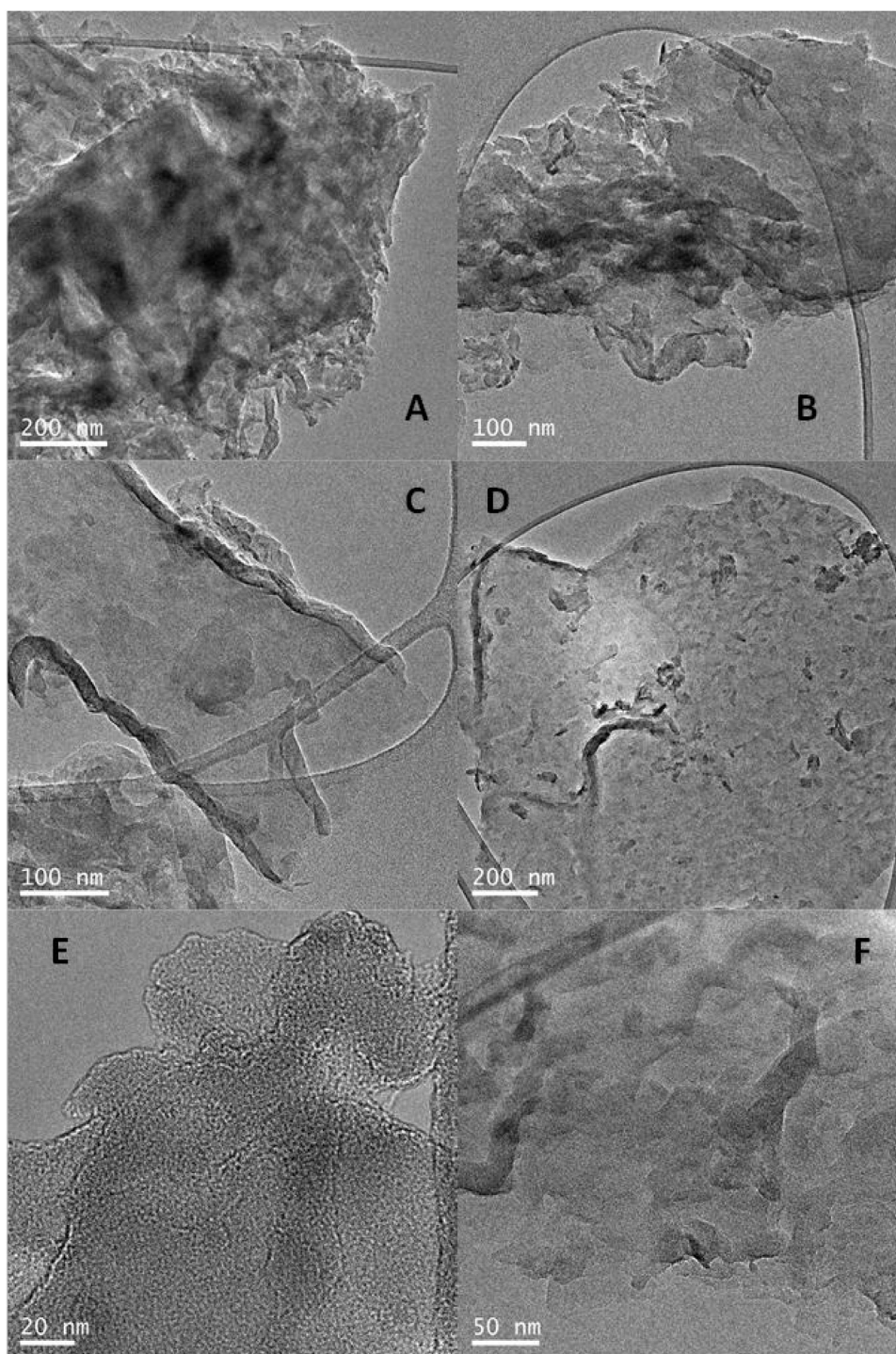
S4) [67]. This mixture has been already detected in similar samples [39]. Here we observed rather small contributions of the most oxidized Mn (IV) species while the other two chemical states dominate the spectra. Most important, absence of significant differences is noted between the Mn chemical state when supported on bulk or the exfoliated materials. The differences in the carbon nitride component do not affect the Mn oxidation state. This is a consequence of the similar interaction as well as Mn dispersion. In fact, the Mn/(C+N) atomic ratio determined by XPS show limited differences (0.014 for g-C<sub>3</sub>N<sub>4</sub> and 0.012 for E-g-C<sub>3</sub>N<sub>4</sub>).

To end up the physico-chemical characterization of the catalysts, morphological differences between the exfoliated and bulk components are further investigated through a TEM study (Fig. 7). The A and B micrographs correspond to the g-C<sub>3</sub>N<sub>4</sub> and g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub> samples, respectively. Both display several darker areas caused by the stacking of a significant number of carbon nitride sheets. On the other hand, the E-g-C<sub>3</sub>N<sub>4</sub> and E-g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub> catalysts (C and D) show a nanosheet-type structure, with significant less interference or shadowing effects coming from the stacking of carbon nitride layers. The differences in the stacking of layers between the bulk and the exfoliated specimens are also evident in the low magnification SEM images presented at the Supporting information section (Fig. S4). The presence of the Mn is not inferred from the microscopy study.

In summary all structural (XRD, microscopy, infrared) and electronic (UV–vis, XPS) studies showed up physico-chemical differences, all of them (but particularly XRD, TEM and UV–vis) consistent with the limitation of the stacking dimension of CN-containing layers in the materials. Of prime importance is that they produced differences in band gap energy and thus can have significance in light absorption and charge handling properties and thus in photocatalytic properties. Concerning the manganese species, XPS results did not detect differences between the chemical state of this component in the bulk and exfoliated samples. A rather mild variation in the dispersion of Mn is however encountered, with a modest increase in the case of the exfoliated vs. the bulk material.

### 3.2. Photocatalytic elimination of toluene

The photochemical performance of the obtained materials in elimination of toluene under UV, visible, and sunlight-type illumination is reported in Fig. 8. The reaction rate is maximized under UV excitation but relatively close (although inferior considering the error of the measurement) values are obtained upon sunlight-type excitation. Significantly lower values are encountered while using exclusively visible light excitation. The exfoliated samples are clearly superior to the bulk ones and the presence of MnO<sub>x</sub> seems to affect activity as a function of the nature (bulk or exfoliated) of the carbon nitride main component. Compared with the P25 reference sample indicates relatively low rates under UV and Sunlight-type but, on the contrary, the carbon nitride samples display a significant increase on the reaction rate under visible light illumination due to the almost null performance of P25 in such conditions [68–70]. Note that P25/Titania catalysts works through a hydroxyl or hole-related mechanism [33], while the carbon nitride samples are only able to produce (at least initially) electron-related charged species [20,71],



**Fig. 7.** TEM views of (A) bulk g-C<sub>3</sub>N<sub>4</sub>, (B) bulk g-C<sub>3</sub>N<sub>4</sub> modified with MnO<sub>x</sub>, (C) and (E) g-C<sub>3</sub>N<sub>4</sub> exfoliated, and (D) and (F) g-C<sub>3</sub>N<sub>4</sub> exfoliated modified with MnO<sub>x</sub>.

indicating that the comparison of both types of samples is complex. In any case, comparison with Ag-promoted carbon nitride materials shows a less significant effect of MnO<sub>x</sub> for a bulk carbon nitride material having a similar loading that here tested [72]. MnO<sub>x</sub> surface species have however a different behavior in the exfoliated material. Samples differ in activity (Fig. 8), as already discussed in this paragraph, but also in selectivity (Table 2). Selectivity differences are mostly encountered in presence/absence of MnO<sub>x</sub> species as well as with the nature of the illumination source. The latter is relatively modest and any case minor with respect to the first.

**Table 2**  
Reaction selectivity of toluene photodegradation.<sup>a</sup>

Sample	UV		Sunlight-type		Visible	
	Bz <sup>b</sup>	CO <sub>2</sub>	Bz <sup>b</sup>	CO <sub>2</sub>	Bz <sup>b</sup>	CO <sub>2</sub>
g-C <sub>3</sub> N <sub>4</sub>	46	54	41	59	58	42
E-g-C <sub>3</sub> N <sub>4</sub>	42	58	44	56	49	51
g-C <sub>3</sub> N <sub>4</sub> /MnO <sub>x</sub>	56	44	58	42	60	40
E-g-C <sub>3</sub> N <sub>4</sub> /MnO <sub>x</sub>	51	49	61	39	63	37

<sup>a</sup> Average Standard error: 4.3%.

<sup>b</sup> Bz: benzaldehyde.

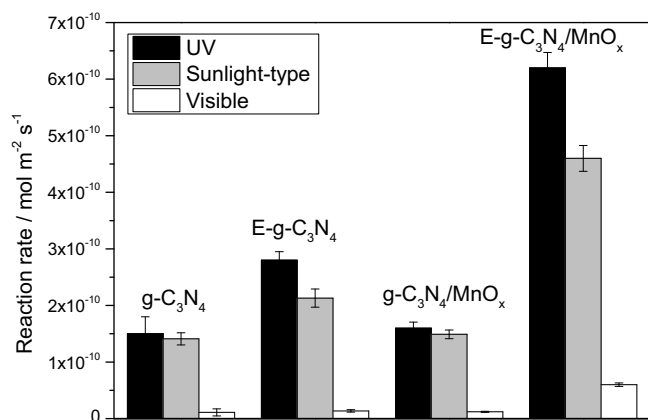


Fig. 8. Surface area normalized reaction rate for the carbon nitride based samples.

To compare the sample performance on quantitative basis the calculation of the true quantum efficiency must be carried out. This is here computed following the recipes presented in Section 2.4.2, which require the local superficial rate of photon absorption,  $e^{a,s}$ , calculation as well as the measurement of the selectivity of the reaction. The obtention of the  $e^{a,s}$  observable requires, according to Figs. 1 and 2, calculation on a cylindrical geometry and thus two independent geometrical variables. The local superficial rate of photon absorption observable across the corresponding two dimension reaction coordinates are presented in Fig. 9 under all (UV, visible and sunlight-type) illumination conditions tested in the study. The panels of Fig. 9 indicate the same geometrical behavior for all samples under study and thus differences between samples  $e^{a,s}$  can be represented by a single number, the average value of  $e^{a,s}$ . Using thus the average values of data presented in Fig. 9, the selectivity information of Table 2 and Eq. (12) we computed the true quantum efficiency values under all illumination conditions tested.

Fig. 10 provides a graphical representation of the true quantum efficiency values obtained in the toluene photo-degradation using our carbon nitride containing catalysts. The comparison of the effi-

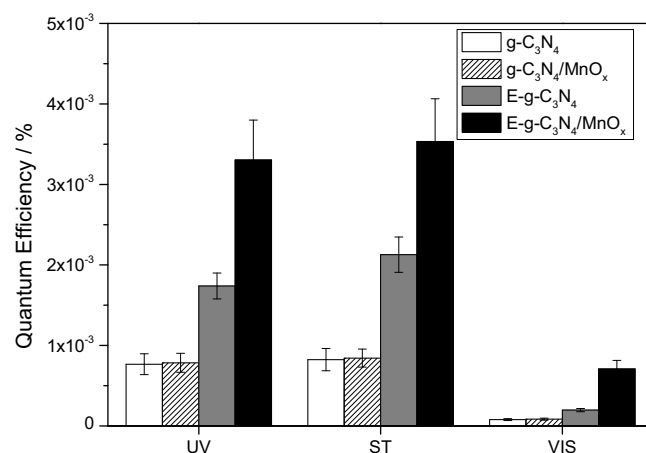


Fig. 10. Quantum efficiency for the carbon nitride based samples.

ciency values indicates the relatively high performance achieved under sunlight-type and UV excitation while significantly lower values are presented under visible light. Maximum values for all catalysts are obtained under sunlight-type excitation. This contrast (although differences are modest) with the reaction rate, displaying in the last case maximum values under UV.

The comparison of catalysts performance is carried out in Table 3. In this table and by providing the ratio between the efficiency parameter of bulk and exfoliated samples as well as in presence or absence of MnO<sub>x</sub> we attempt to measure in quantitative basis the influence of the exfoliation or the surface modification with Mn of the carbon nitride material. The surface deposition of MnO<sub>x</sub> rendered, as mentioned, rather similar species in terms of their physico-chemical variables for both bulk and exfoliated samples but the catalytic effect as measured in Table 3 is different. The comparison of Mn-containing bulk and exfoliated carbon nitride materials vs. their bare counterparts (first two rows of Table 3) indicates an almost ×2 enhancement in the exfoliated case under UV and Sunlight-type illumination and of more than three times in the case of visible light. No significant differences are encoun-

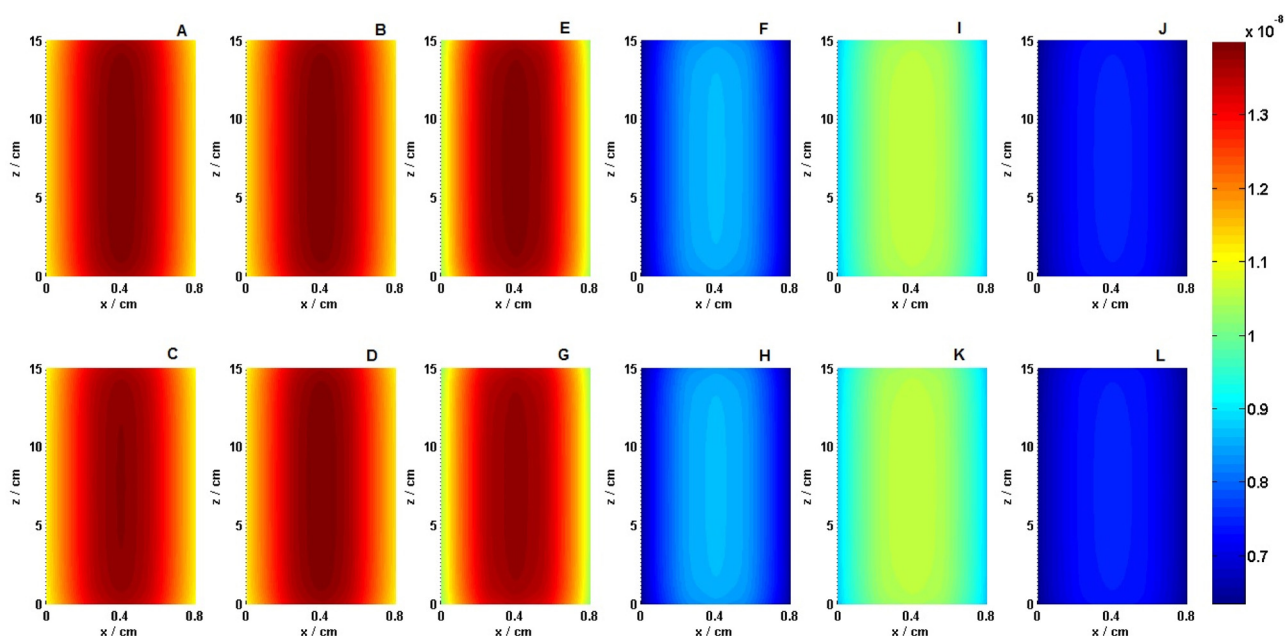


Fig. 9. Local superficial rate of photon absorption (Einstein cm<sup>-2</sup> s<sup>-1</sup>) of the samples under the different irradiation conditions (UV, sunlight-type and visible). The presentation of the panel follow this sequence: g-C<sub>3</sub>N<sub>4</sub>, E-g-C<sub>3</sub>N<sub>4</sub>, g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub> and E-g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub>. From A to D (UV). From E to F (Sunlight-type). From I to L (Visible).



**Table 3**

Enhancement ratio produced by the presence of MnO<sub>x</sub> species or the exfoliation process in toluene photodegradation quantum efficiency by carbon nitride samples.<sup>a</sup>

True Quantum Efficiency Ratio	UV	Sunlight-type	Visible
MnO <sub>x</sub> Effect			
g-C <sub>3</sub> N <sub>4</sub> /MnO <sub>x</sub> vs. g-C <sub>3</sub> N <sub>4</sub>	1.0	1.0	1.1
E-g-C <sub>3</sub> N <sub>4</sub> /MnO <sub>x</sub> vs. E-g-C <sub>3</sub> N <sub>4</sub>	1.9	1.7	3.6
Exfoliation Effect			
E-g-C <sub>3</sub> N <sub>4</sub> vs. g-C <sub>3</sub> N <sub>4</sub>	2.3	2.6	2.5
E-g-C <sub>3</sub> N <sub>4</sub> /MnO <sub>x</sub> vs. g-C <sub>3</sub> N <sub>4</sub> /MnO <sub>x</sub>	4.2	4.2	8.4

<sup>a</sup> Average standard error: 31%.

tered in the case of the bulk carbon nitride material. The differential catalytic effect of Mn in bulk-type and exfoliated carbon nitride materials has not obvious roots as Mn oxidation and dispersion state seem rather similar in both types of samples. A more efficient electronic interaction with the exfoliated support by effect of electronic confinement may be envisaged but requires further study to be substantiated.

The catalytic effect of the exfoliation process is measured and presented in the last two rows of Table 3. The enhancement ratio is always positive and larger than the effect ascribable to the presence of MnO<sub>x</sub> species in the bulk or exfoliated carbon nitride materials. An increase of the efficiency related to the exfoliation process of ca of 2.5 times is observed for all illumination conditions in absence of Mn. In presence of Mn, we encountered ca. ×2 enhancement with respect to the one presented in absence of such chemical species under UV and Sunlight-type illumination conditions and a further ×4 in the case of Visible illumination. As a result of the carbon nitride dual modification, we observe an efficiency increase of ca. ×4 (UV and Sunlight-type) or ×8 (Visible) with respect to the bare carbon nitride sample. The exfoliation effect in photo-activity can be ascribed to the above described confinement effect in the electronic properties of the carbon nitride material.

From the previous numbers we can conclude that a consistent, positive (enhancement) effect induced by both the presence of surface MnO<sub>x</sub> species and the exfoliation process. Quantitative measure (as presented in Table 3) of both modifications of the carbon nitride material indicates the similar (magnitude) catalytic effects produced under UV and Sunlight-type and a more important enhancement taking place under visible illumination. This quantitative estimation indicates that the energy of the charge carriers after light absorption influences strongly photo-activity, being the effect maximized under Sunlight (Fig. 10) by an effect mostly ascribable to UV photons, but the estimation also shows that the enhancement effect due to the presence of MnO<sub>x</sub> presence or the exfoliation process have a more pronounced positive effect for visible rather than UV photons. However, visible photons do not generate chemistry to the level of UV photons. Table 3, on the other hand, does not provide a complete answer as to measure and understand the effect of these two modifications of the bulk carbon nitride material. Although the effect of MnO<sub>x</sub> is barely observed in the case of the bulk materials and always of lower magnitude (roughly two times) than the one of the exfoliation process, the comparison of enhancement values of the two first as well as the two last columns of Table 3 indicates that, in addition to their “isolated”, positive effects in the original bulk carbon nitride material, there is a synergistic effect between the two physical modifications of the bulk carbon nitride material.

To measure on quantitative basis the synergistic effect using the true efficiency parameter we present in Table 4 an excess (or synergy) function. A simple function of the henceforth called ideal, ternary system (where the effects of both physical modifications of the bulk carbon nitride materials are simultaneously present) can

**Table 4**

Excess function (defined by the ratio between the observable measured for sample E-g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub> and Eq. (14)) produced by the simultaneous presence of surface MnO<sub>x</sub> species and the exfoliation process of carbon nitride in toluene photodegradation quantum efficiency.<sup>a</sup>

Measurement	UV	Sunlight-type	Visible
E-g-C <sub>3</sub> N <sub>4</sub> /MnO <sub>x</sub> Quantum Efficiency	$3.3 \times 10^{-3}$	$3.5 \times 10^{-3}$	$7.1 \times 10^{-4}$
Eq. (14)	$1.7 \times 10^{-3}$	$1.95 \times 10^{-3}$	$2.0 \times 10^{-4}$
Excess Function	1.9	1.8	3.5 <sub>5</sub>

<sup>a</sup> Average standard error: Quantum efficiency 20.5%; Excess Function 26%.

be defined with respect to the main reference (bulk carbon nitride) and the parent binary (i.e. having a single modification) systems [33]:

$$\eta_q(E/MnO_x) = \eta_q(g - C_3N_4) + (\eta_q(E - g - C_3N_4) - \eta_q(g - C_3N_4)) + (\eta_q(g - C_3N_4/MnO_x) - \eta_q(g - C_3N_4)) \quad (14)$$

Eq. (14) is defined to provide a way to measuring quantitatively if the physical modification triggered jointly by the exfoliation (right-hand side second term of the equation) or the presence of surface MnO<sub>x</sub> species (right-hand side third term of the equation) in the bulk g-C<sub>3</sub>N<sub>4</sub> catalyst is detrimental, positive but simply additive or, as expected in a good photocatalyst, synergistic. The ratio between the real, measured efficiency for the E-g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub> sample and that corresponding to the mentioned ideal, ternary system  $\eta_q(E/MnO_x)$ ; ideal) corresponds to an “excess” function which can be lower, equal or higher than 1. The set of the above mentioned observables concerning the E-g-C<sub>3</sub>N<sub>4</sub>/MnO<sub>x</sub> sample efficiency, the outcome of Eq. (14) and the excess ratio are presented in Table 4.

Inspection of the first two rows clearly indicates that combination of the two physical modifications of the bulk carbon nitride material generates a significant synergistic catalytic effect as measured through the true quantum efficiency. The third row quantifies the synergy in catalytic terms, indicating that the combined action of surface MnO<sub>x</sub> species and the exfoliation has a roughly ×2 (UV and Sunlight-type) or ×3.5 (Visible) enhancement with respect to the isolated effect generated by each one. This therefore provides a quantitative measurement of the catalytic effects played by both modifications of the carbon nitride component indicating that the synergy between them is wavelength dependent but always significant (2–4 times).

#### 4. Conclusions

In this work we synthesized a bulk-type, relatively low surface area carbon nitride material through a well-established procedure consisting on the calcination of a melamine precursor in a semi-closed system. The obtained semiconductor was subjected to physical modification related to a sono-assisted exfoliation process in water as well as to a surface deposition of Mn-containing species through an impregnation process. The physico-chemical properties of the resulting photocatalysts were measured and analyzed using a set of characterization techniques. The overall analysis of these studies indicates that the exfoliation treatment is effective in producing a less packed material having physico-chemical (particularly evidenced in the XRD, TEM and UV–vis data) properties affected by confinements affects derived from a delamination process. The study also showed that Mn species supported in the carbon nitride materials do not display significant differences in terms of the oxidation state and dispersion between the exfoliated and bulk carbon nitride samples.

Through the measurement of the optical properties of the system (materials and reactor constituents) and mathematical modeling we computed the true quantum efficiency for the photo-



degradation of toluene. Values of the true quantum efficiency allow to quantitatively estimating the catalytic effect of the two physical modifications suffered by the initial, bulk-type carbon nitride semiconductor. Although the two modifications are positive (or non-negative), we observe that the exfoliation process promotes activity about two times more efficiency than the surface Mn-containing species irrespective of the photon wavelength, i.e. for all UV and visible light photons, or the surface area of the carbon nitride material. Moreover, a strong synergistic effect, again two times with respect to the sum of the “isolated” components, is also observed when carbon nitride is subjected simultaneously to both modifications. The procedure here outlined for the analysis of the true quantum efficiency parameter provides a simple route to quantitative estimate and interpret the catalytic effects originated by physical modifications of carbon nitride materials and can be generalized to all photocatalytic materials and tests for composite, ternary materials.

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## Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at <http://dx.doi.org/10.1016/j.apcatb.2016.10.044>.

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